First Principles Investigation of the Influence of Varied Cr Atom on Band Structure and Magnetic Moment of Rutile SnO₂

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Abstract: The electronic and magnetic properties of SnO_2 doped with various compositions of chromium atoms are reported. Studies on magnetic property of $Cr_xSn_{1-x}O_2$ compounds features possible room temperature ferromagnetism which increased nonlinearly as Sn atom is replaced with Cr atom, at $x = 0.25(1.9976 \,\mu\text{B}), x = 0.50(3.9309 \,\mu\text{B}), x$ $= 0.75(5.8831 \,\mu\text{B})$ and $x = 1.00(7.821 \,\mu\text{B})$. The magnetic moment and bandgap energy of undiluted SnO_2 were compared at x = 0. The addition of Cr atom into SnO_2 enhanced the shift from pure binary nonmetallic system to ternary metallic compound. The direct energy gaps decrease from x = 0 to 0.5, and increase from x = 0.75 to 1.00.

Keywords: *Rutile, First-principles, Generalized Gradient Approximation (GGA), Magnetic moment, magnetic dipole moment.*

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1.0 Introduction

Dilute magnetic semiconductors doped with transition metals are currently receiving research interest because of their functionalities in the production of spin transistors as well as lightemitting diodes (Ogale, 2010). Stannic oxide

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(SnO₂) is a wide-band n-type semiconductor material and it has application potential in photo decadence of organic substances, catalysts, coatings, optoelectronic devices and resistors, opacifiers, photosynthesis, etc.

Electrons in an atom perform orbital and spin motions which are capable of producing a magnetic field. Strong or weak magnetism in materials; is a function of magnetic moments produced from the amperian currents and can be segmented into: ferromagnetism, diamagnetism, paramagnetism, antiferromagnetism and ferrimagnetism (Okeke *et al.*, 2000). Ferromagnetic materials exist in domains with varying magnetic moments. In the presence of an externally applied magnetic field, the aligned magnetic moments within domains become magnetized with a very strong internal magnetic field (Okeke *et al.*, 2000).

Ferromagnetism tends to diminish with an increase in temperature but gradually vanishes as the Curie temperature is reached. A Ferromagnetic material comprises of soft materials, hard materials and ferrites and they have permeability greater than one. These materials are useful in the production of permanent magnets, transformer core, magnetic tapes and memory storage devices and some components in the automobile, electronic sector, communication and aviation industries. They are also useful in realizing magnetic resonance imaging and magnetic screening.

Ternary and quarternary diluted magnetic semiconductor materials are transition metal-doped systems that reveal the room temperature ferromagnetism (Fukumura *et al.*, 2005). Some studies have been reported on first principles electronic calculations concerning rutile SnO₂. For example, Borges *et al.* (2011) carried out such calculations within the spin density functional theory, alternated magnetic and nonmagnetic layers of rutile-CrO₂ and rutile SnO₂. They observed a half-filled for the (CrO₂)_n(SnO₂)nSL_s at all values of n. The ground state was found to be Ferromagnetic (FM) with a magnetic moment of 2

 μ B. Abdulsattar *et al.*(2016) reported effects of Indium doped tin oxide by both experimental and theoretical approaches. Their experimental results through XRD, UV-Vis and Raman spectroscopy reveal that as lattice constants increased, there was a decrease in energy gap. The DFT method via

Large unit cell showed a change in experimental stoichiometry as the lattice constant before and after the doping are almost the same.

In view of the significant of the magnetic and electronic structure of rutile SnO_2 on the potential applications of previous works, the present study is aimed at investigating the influence of varied Cr atom on band structure and magnetic moment of rutile SnO_2

2. 0 Computational Details

All computational calculations were performed using the Perdew- Burke – Ernzhof - Projected Augmented Wave (PBE-PAW) pseudopotentials from the Quantum espresso library (Gianozzi *et al.*, 2009). A six atom supercell was built for an undiluted binary SnO_2 at x = 0. Twelve atoms supercell comprising of four atoms of tin and eight atoms of oxygen were employed at x = 0.25 to 1 with Cr atom replacing Sn atom at x = 0.25 to 1. A Monkhorst- Pack k-point of 10x10x5 was applied for SnO_2 at x = 0, with an energy, cut-off of 80 Ry. At x = 0.25 to 1, for a twelve atom supercell used, the k-points were set at 10x10x5, 8x8x4, 10x10x5and 10x10x5 with an energy cut off of 90Ry.

2.1 Methodology

The magnetic moment (m) depicts the magnetic strength and magnetic orientation which brings about a magnetic field. It aligns with an external applied magnetic field and links torque (τ) of an object such as permanent magnets, astronomical objects, elementary particles and loops of electric current to the magnetic field (B). This can be expressed mathematically as:

$$\tau = m \times \mathbf{B} \tag{1}$$

The magnetic field of a magnetic dipole is proportional to its magnetic dipole moment (μ) . The magnetic dipole is a measure of a dipole's ability to turn itself into alignment with a given external magnetic field. The magnitude of energy (E) of magnetic dipole moment (μ) is a function of the alignment of magnetic field (B). That is,

$$\mathbf{E} = -\ \boldsymbol{\mu} \times \mathbf{B} \tag{2}$$

Electrons possess inherent magnetic moments which are related to their spin angular momentum. In the presence of a magnetic field, an electron has one of two orientations in agreement with the magnetic spin quantum number. And the spin polarization is the ratio of the density of states upspin to down spin electrons at a Fermi level (Yasuhiko *et al.*, 2011). The polarization of the electronic spins leads to magnetization density because magnetization is a function of the ground state charge density and any other ground state property. The magnetic dipole moment per unit volume is:

$$M(r) = \sum_{i} N_i \langle m_i \rangle$$
 (3)

where $\langle m_i \rangle$, the average magnetic dipole moment of the ith is the type of molecule in the vicinity of point r and N_i is the number of such molecules per unit volume at r.

The energy gap between the uppermost valence band and the minimum conduction band differs for various materials. It decides the electrical conductivity and electronic transport of a solid, and classifies materials into insulators, semiconductors and conductors. Energy gaps due to varying concentrations of chromium atom into SnO_2 were examined using a modified form of Vegard's law (Ayedun *et al.*, 2017):

$$E_{g}(A_{x}B_{1-x}C) = xE_{AC} + (1-x)E_{BC} - x(1-x)b_{E}$$
(4)

3.0 Results and Discussion 3.1 Band structural property

The band structures along the high symmetry paths were observed in this study at x = 0, 0.25, 0.50, 0.75 and 1.00 as shown in Figs. 1 to 5 respectively. A bulk binary SnO₂ at x = 0 is characterized with a direct bandgap; which is feasible at Γ to Γ with direct energy of 1.163eV. The energy gap obtained is closer to the energy gap of 1.25eV reported by Deligoz *et al.* (Deligoz, 2007). Also, indirect energy gaps were achieved from M to Γ at x =0.25, Z to Γ at x = 0.75 and Γ to X at x = 1.00respectively while the direct energy gaps was attained from Γ to Γ at x = 0.5. The energy gaps varied nonlinearly as the dopant, Cr atom increased. The energy gaps were computed using equation (4). The most stable phase of the material, $Cr_xSn_{1-x}O_2$ is at x = 0.25 with the bowing energy parameter of 0.385eV. This will be of immense value in the production of spintronic devices and spin-polarized light-emitting diodes. There is a hybridization between O-2p state at the topmost valence band of SnO₂ and minimum conduction, Sn-5p state at x = 0. For the compositions x = 0.25

to 0.75, the uppermost valence band was observed at O-2p state while the conduction bands were predominated by Sn-5p state and Cr- 3d state.

At x = 1, the lowest conduction band was dominated by Cr- 3d state and the topmost valence band by O-2p state. The values of band energies of $Cr_xSn_{1-x}O_2$ obtained in this study are reported in Table 1.

Compositions (x)	System	Present Study (eV)	Experimental(eV)	Theoretical
0.00	SnO_2	1.163	3.61[4]	1.25 [15]
0.25	$Cr_{0.25}Sn_{0.75}O_2$	1.000		
0.50	$Cr_{0.50}Sn_{0.50}O_2$	0.389		
0.75	$Cr_{0.75}Sn_{0.25}O_2$	0.634		
1.00	CrO ₂	0.800	0.00	1.34[5], .5[16]





Fig. 1: Band structure of SnO₂.









Fig. 3: Band structure of Cr_{0.50}Sn_{0.50}O₂.





Fig. 5: Band structure of CrO₂.

3.2 Magnetic Property

The magnetic moment of paired Sn^{2+} and O^{2-} in pure SnO_2 was found to be zero, which is in agreement with the observations made by Wang (2010), Gu *et al.* (2008). SnO_2 doped with Cr atom exhibited a shift in transition from nonmetal to ferromagnetic materials at x = 0.25 to 1. The resultant magnetic moment increased nonlinearly with a rise in composition of Cr atom and their unpaired valence electrons were: Sn $(4d^{10}5s^25p^2)$, Cr $(3S^23p^63d^4)$ and O $(2s^22p^4)$.



The bar chart in Fig. 6 depict the nonlinear increase more in the magnetic moment of $Cr_xSn_{1-x}O_2$ materials Tab

with varied Cr atom while detailed magnetic

moments of the system under study is presented in Table 2.

Compositions (x)	System	Present Study(μB)	Experimental	Theoretical
0.00	SnO ₂	0	0 [19]	0 [11]
0.25	$Cr_{0.25}Sn_{0.75}O_2$	1.9976		
0.50	$Cr_{0.50}Sn_{0.50}O_2$	3.9309		
0.75	$Cr_{0.75}Sn_{0.25}O_2$	5.8831		
1.00	CrO ₂	7.8271		9.0 [9]

Table 2: Magnetic moment of Cr_xSn_{1-x}O₂ materials





4.0 Conclusion

Self-consistent PBE-PAW calculations using GGA approach between exchange correlations based on density functional theory were carried out on $Cr_xSn_{1-x}O_2$ materials. The results obtained indicated that the showed excellent agreement between the experimental and theoretical magnetic moment of pure SnO₂. The most stable phase of these materials was revealed at x = 0.25. The energy gap is in close agreement with theoretical data and 0.67 per cent error with that of experimental.

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Conflict of interest

The authors declared no conflict of interest

