

# First Principles Calculations of Structural, Electronic and Optical Properties of Nitrogen-Doped Titanium Dioxide for Solar Cells Application

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**Abstract:** The dire requirement for less toxic, eco-friendly, cheaper, cost-effective and efficient material for solar cell application has led to increasing focus on a range of different source materials. In particular, the larger energy bandgap in  $TiO_2$  has limited its application for solar cell applications. However, doping  $TiO_2$  with non-metal such as N gives a broader absorption at the visible region and subsequently adjusts the bandgap, which allows better utilization of the solar spectrum. However, to exploit its potentials, a detailed analysis of structural, electronic and optical properties of N doped  $TiO_2$  is necessary. In this work, first-principles calculations within the density functional theory (DFT) are carried out as an approach to address the problem. The calculated bandgap energy for pure  $TiO_2$  (2.30 eV) was in strong agreement with the experimental value. The substitution of nitrogen (N) atom in the  $TiO_2$  at the oxygen (O) and titanium (Ti) sites led to the reduction in the energy gap and the observation was also in good agreement with results from previous works. Our findings confirmed that non-metal doping narrows the energy band gap of semiconductor materials. The optical gap of 1.63 and 0.32 eV for N doped  $TiO_2$  at oxygen (O) and titanium (Ti) sites, which indicated that N-doped  $TiO_2$  can be used to detect light in the near infrared and visible light regions. Direct energy gap, narrowing effects and strong light absorption of N-doped  $TiO_2$  in the near infrared to visible light region suggest that the investigated material is most likely suitable for

solar cells and near infrared optoelectronic applications.

**Keywords:**  $TiO_2$ , DFT, doping, nitrogen, solar cell.

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## 1.0 Introduction

Several investigations on properties, behaviour and applications of transitional metal oxides have been carried out to create satisfaction for the increasing and demanding needs of

communities and industries (Maduraiveeran *et al.*, 2019, Han *et al.*, 2020). The main demand of the industry is to find economical alter environmentally which should be non-toxic, environmentally friendly, easily accessible and efficient (Dixit, 2012, Dash *et al.*, 2018). Consequently, numerous experimental and theoretical investigations are ongoing, in a search for better alternatives for the existing ones. Titanium dioxide ( $\text{TiO}_2$ ) is one of the essential transition metal oxide semiconductors that has given a significant boost to several industries, including the green energy industries. Numerous properties of  $\text{TiO}_2$  have been explored towards the achievement of its applications, notably, its high optical transmittance, strong absorbance of the visible and ultraviolet lights, low electrical resistivity, high ionic semiconductor (Mikami *et al.*, 2000). In addition, the characters that enable its applications in flat panel displays, sensor, pigment, catalyst and transparent optoelectronic devices have also encouraged further research on the applications of  $\text{TiO}_2$  (Yang *et al.*, 2014, Dixit, 2012, Wang *et al.*, 2012, Hitosugi *et al.*, 2010, Wagemaker *et al.*, 2002)  $\text{TiO}_2$  exists in several polymorphs of rutile, anatase, brookite, columbite, baddeleyite, cotunnite, pyrite and fluorite. Three of them exist naturally while others need to be engineered. Among these, the three naturally occurring polymorphs of rutile, anatase and brookite have been synthesized extensively due to their remarkable properties (Wang *et al.*, 2012, Thilagam *et al.*, 2011, Sai and Bang-Gui, 2012). Rutile structure possesses the highest stability (Jaćimović *et al.*, 2010) compare to others, anatase (Beltran' *et al.*, 2001). However, in solar cell technology,  $\text{TiO}_2$  has limited applications due to its wider bandgap nature. To use  $\text{TiO}_2$  for a solar cell application, its bandgap should be tuned to capture the maximum range of the solar spectrum. Currently, doping is one of the best options for the tuning and improvement of the optoelectronic properties of some materials suitable for applications as absorbing layer for

solar cells (Cerdán-Pasarán *et al.*, 2019, Jiang *et al.*, 2019). Doping can significantly adjust the physical properties of materials (Cerdán-Pasarán *et al.*, 2019). Consequently,  $\text{TiO}_2$  is doped repeatedly with diverse transition elements and metals to enhance its optical absorption, magnetic properties, conductivity as well as tune its bandgap (Tseng *et al.*, 2016, Li *et al.*, 2019, Avram *et al.*, 2021, Piątkowska *et al.*, 2021). In comparison with metal doping, doping of non-metals is an effective approach for narrowing the band-gap of material and thus lowering the energy requires for the electronic transition from the HOMO to the LUMO (Reisner and Pradeep, 2014). Non-metal doping with an atom such as N can enhance the optical absorption, conductivity, magnetic and electronic properties of semiconductor materials (Duan *et al.*, 2015, Asahi *et al.*, 2001, Wang *et al.*, 2019).

Given the expected benefits of doping  $\text{TiO}_2$  with nitrogen, the present study is aimed at implementing a theoretical investigation of the effect of doping N on structural, electronic, and optical absorptions of  $\text{TiO}_2$  semiconductors. This was accomplished by adopting a first-principles approach based on DFT implemented within the Quantum Espresso package. Different exchange-correlation functional of GGAs and LDA were employed as the manipulated variables in this study.

## 2.0 Computational Details

The calculations presented in this work were performed with two open-source simulation packages, Quantum Espresso (Giannozzi *et al.*, 2009) and Yambo (Marini *et al.*, 2009) codes. Yambo simulation package interfaces with QE for excited-state calculations (which uses DFT results as its input) were also employed for the calculations. (Sangalli *et al.*, 2019, Barhoumi and Said, 2020, Marsili *et al.*, 2016, Lawal *et al.*, 2018) Electronic structure and optical properties of pure and N doped  $\text{TiO}_2$  calculations were performed based on the pseudopotential plane-wave method. For N doped  $\text{TiO}_2$ , one atom of oxygen and titanium atoms were replaced by an



atom of N for building the doped structures. Figs. 1(a)-(c) show the three models. Wu-Cohen's generalized gradient approximation (WC-GGA) (Perdew *et al.*, 1996a) Perdew-Burke-Ernzerhof generalized gradient approximation (PBE-GGA) (Perdew *et al.*, 1996b) and local density approximation (LDA) (Becke, 1988) exchange-correlation potential were used for lattice dynamics of pure and N doped TiO<sub>2</sub>. Norm-conserving pseudopotentials were used in modeling interactions between ionic core potential and electrons of Ti, O and N atoms. The Brillouin zone integration was performed using Monkhorst-Pack grid (Monkhorst and Pack, 1976) of 11 × 10 × 11 k-points grids for structural and band structure calculations for both undoped and doped systems. 70 Ry cut-off energy of the plane-wave basis set was used to expand the wave functions and 270 Ry for charge density. However, in the performance of optical properties calculations, the complex dielectric constant was first evaluated, and the other properties were expressed in terms of it. The complex dielectric constant  $\epsilon$  is given by  $\epsilon(\omega) = \epsilon_1(\omega) + i\epsilon_2(\omega)$ , where  $\epsilon_1$  and  $\epsilon_2$  are real and imaginary parts of dielectric constant respectively. This suggest that, the complex dielectric function is suitable for the description of the optical properties at

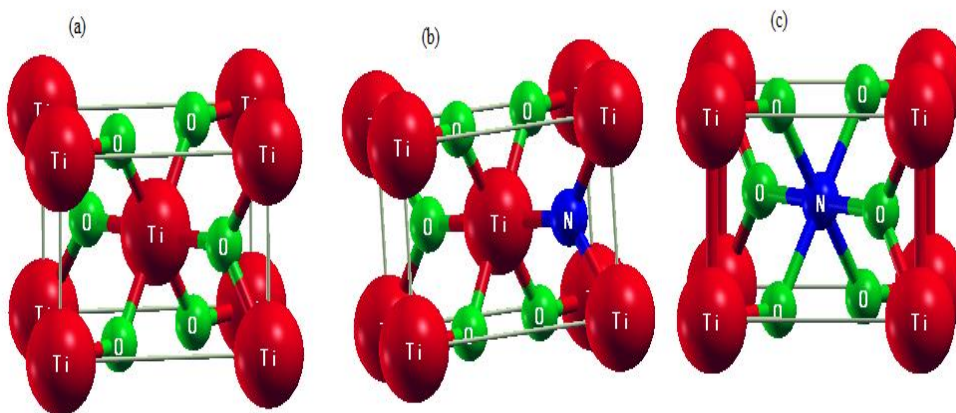
microscopic level or quantum mechanical level. The expression for the complex imaginary part of dielectric function  $\epsilon_2(\omega)$  has been derived in equation 1.

$$\epsilon_2(\omega) = \frac{2\pi e^2}{\Omega \epsilon_0} |\langle \psi_k^c | \hat{u} \times \vec{r} | \psi_k^v \rangle | \delta(E_k^c - (E_k^v + E)) \quad (1)$$

### 3.0 Results and Discussions

#### 3.1 Structural properties of TiO<sub>2</sub>

TiO<sub>2</sub> naturally occurs in three structures, namely, rutile, anatase and brookite. In this work, the rutile crystal structure of TiO<sub>2</sub> was optimized using the Quantum Espresso package and the optimised structure was shown in Fig. 1. The optimization was carried out using different DFT functionals, including GGA-PBE, GGA-PBESol, GGA-WC and LDA. By exploiting the Murnaghan's equation of state, the structural parameters of lattice parameters are obtained. The results of the structural parameters are tabulated in Table 1. Generally, our calculated lattice parameters (*a*, *b* and *c*) are in good agreement with experimental work, with the difference only up to 3.0% and 2.8% respectively. Furthermore, the calculated lattice parameters with GGA-PBE were found to be closer to experimental results compared to GGA-PBESol, GGA-WC and LDA.



**Fig. 1.** Crystal structure of (a) Pure TiO<sub>2</sub> (b) N Doped TiO<sub>2</sub> at O (c) N doped TiO<sub>2</sub> at Ti site



**Table 1: Calculated and experimental lattice constants of TiO<sub>2</sub>**

Ref.	XC	<i>a</i> (Å)	<i>b</i> (Å)	<i>c</i> (Å)
Present work	LDA	4.634	4.634	2.976
	PBE-GGA	4.543	4.543	2.961
	PBEsol	4.613	4.613	2.849
	GGA-WC	4.640	4.640	2.990
Experiment (Gerward and Olsen, 1997)		4.592	4.592	2.958

### 3.2. Electronic properties of pure TiO<sub>2</sub> and N Doped TiO<sub>2</sub>

Electronic properties calculation is very important for describing the optoelectronic properties of semiconductor material. The electronic properties investigations of TiO<sub>2</sub> and N doped TiO<sub>2</sub> cover the electronic band structure, density of state (DOS) and partial density of state (PDOS). The main purpose of the ground state, electronic band structure, DOS and PDOS calculations in this work is to obtain Kohn Sham (KS) eigenvalue and eigenfunctions as well as useful information about the electronic properties of the concerned materials. The calculated electronic band structures of pure TiO<sub>2</sub> and N doped TiO<sub>2</sub> are displayed in Figure 2. The band structures of both pure and N doped TiO<sub>2</sub> were analyzed along with the special symmetry directions of the irreducible Brillouin zone along with the nine symmetries  $\Gamma \rightarrow X \rightarrow \Delta \rightarrow P \rightarrow \Gamma \rightarrow T \rightarrow Y \rightarrow K \rightarrow H$  directions and the energy range of the band structure is plotted from 0.0 eV to 3.9 eV. The position of the Fermi level in the band structure of these crystals is shown by setting it to be zero on the energy scale. PBE exchange correlation potential is chosen over LDA, because in several cases GGA-PBE gives more reliable and accurate results for DFT electronic properties calculation. Band structures were calculated within PBE plus spin orbit coupling (SOC). For band structure calculations within PBE +SOC, the energy separation between the bottom of the conduction band and the top of the valence band occurred at the  $\Gamma$ , indicating that TiO<sub>2</sub> in the

rutile phase has a direct bandgap with 2.30 eV energy gap, this value is close to experiment value of 3.0 eV when compared with other DFT calculations. Previous band structure calculations of pure TiO<sub>2</sub> in rutile phase within first-principles approach without taking into account the effect of SOC obtained a value of 1.88-2.10 eV [1-5]. The consistency of our result with experiment data is due to the effects of SOC. In order to investigate the effect of N-doping in TiO<sub>2</sub>, calculations were performed for TiO<sub>2</sub> with N substitution. After N-doping at the oxygen site the conduction band edge [Figure 2(b)] shows a slight change in a position towards the lower energy and valence band towards the higher energy. The bandgap of single-atom nitrogen-doped TiO<sub>2</sub> at the oxygen site is reduced from 2.30 to 1.63 eV. It is well known that direct bandgap semiconductor material shows the optical activity of the material, which can be used for optoelectronic and solar cell applications. On the other hand, when the nitrogen atom replaced one atom of Ti atom, the bandgap reduced from 2.30 to 0.32 eV. It can be seen from Figure 2 (c) that the minimum of the conduction band changed from  $\Gamma$  to P point indicating its indirect nature of bandgap. This trend of reduction of bandgap value in single atom doping is in good agreement with the previous theoretical approaches (Butt *et al.*, 2018, Lawal *et al.*, 2021) and experiment measurement (Mushtaq *et al.*, 2016). Our calculated bandgap values together with previous theoretical results and experimental data are given in Table 2.



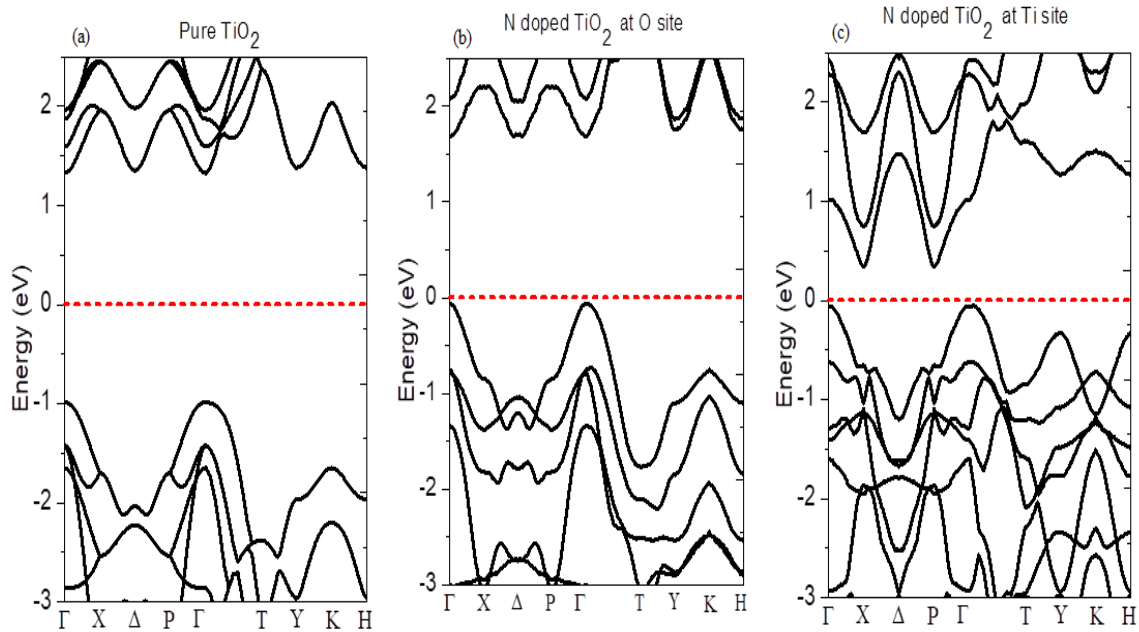
To explore further on the electronic structure of undoped and N-doped TiO<sub>2</sub>, we calculated the density of states (DOS) and partial density of states (PDOS) with DFT+SOC method. The density of states plays an important role in the formation of bands and the study of the bandgap. Results of DOS will assist in clearly understanding the nature of the bands while PDOS provides information about the atomic orbital contribution of the bands. The calculated DOS of undoped and N-doped TiO<sub>2</sub> are displayed in Figure 3. Figure 4 (a) depicts the PDOS of undoped TiO<sub>2</sub> per orbital type that is *s*-, *p*- and *d*-orbitals per Ti and *s*-, *p*- orbital for O atoms. The energetical valence bands which is situated between -1.5 eV to -6.3 eV is principally due to *s*- and *p*-orbitals of Ti and O atoms with a minor contribution of *d*-orbital of Ti atoms. On the other hand, the lowest of the conduction bands consists of *d*-orbitals of Ti atoms with little contribution from *s*- and *p*-

orbitals of Ti and O atoms. PDOS of N-doped TiO<sub>2</sub> by substituting N atom at O sites are shown in Figure 4 (b). The maximum of the valence band near to the Fermi level comes from *p*-orbitals of Ti, *s*- and *p*-orbitals of O and N atoms. Conversely, the *s*-orbital of Ti atom is the main contributor of the lowest valence bands. The minimum conduction bands primarily come from the *p*-orbital of Ti atoms. Figure 4 (c) shows a result of PDOS of N-doped TiO<sub>2</sub> at one Ti site atom. Both valence bands majorly consist of *p*-orbitals of Ti atoms, *s*-orbital of O atoms and *p*-orbital of N atoms with a small contribution from *s*-orbital of Ti atom, *p*-orbitals of O atoms and *s*-orbital of N atoms in terms of hybridization. The conduction band (CB) is composed of *s*-orbital of Ti atoms, *p*-orbitals of O atoms and *s*-orbital of N atoms with little contribution from *p*-orbitals of Ti atoms, *s*-orbital of O atoms and *p*-orbital of N atoms.

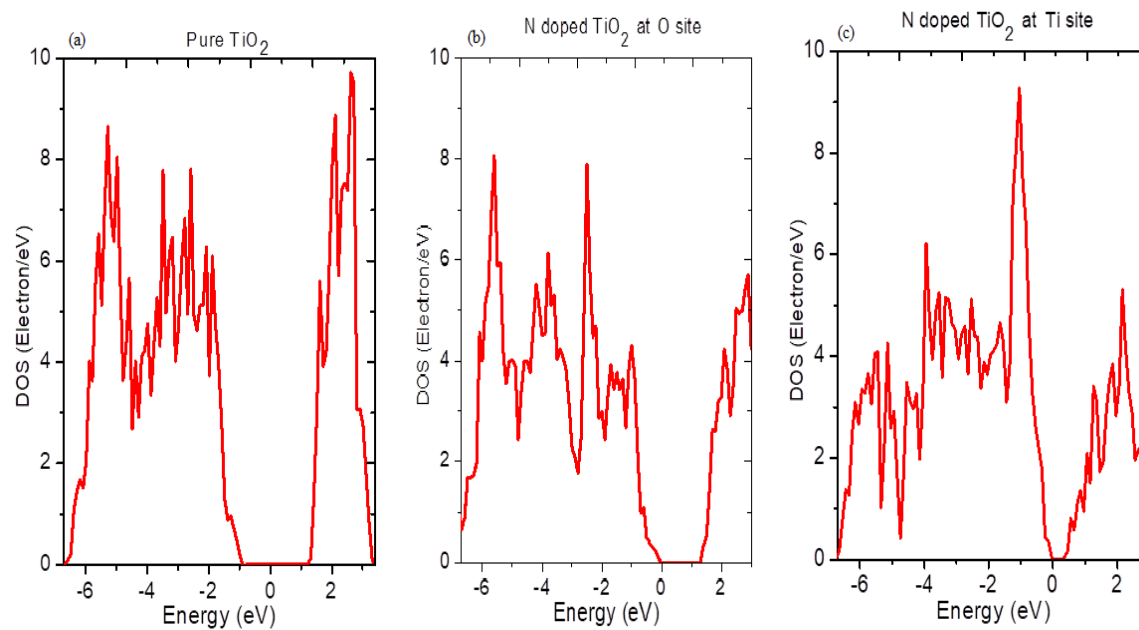
**Table 2: Results of band gap for TiO<sub>2</sub> polymorphs of rutile and N doped TiO<sub>2</sub>. The results are compared with available experimental data and other first principles calculations**

Polymorph	Work	Exchange Correlation Functional	Bandgap value, E <sub>g</sub> (eV)	Type of bandgap
TiO <sub>2</sub> (Rutile)	Present work	PBE-GGA+SOC	2.30	Direct (Γ - Γ)
	Previous work	Experiment(Reyes-Coronado <i>et al.</i> , 2008)	3.00	-
		PBE-GGA(M Landmann <i>et al.</i> , 2012)	1.88	Direct (Γ - Γ)
		PBE-GGA (Sai and Bang-Gui, 2012)	1.89	Direct (Γ - Γ)
		EV-GGA(Baizae and N.Mousavi, 2009)	1.90	Direct (Γ - Γ)
		GGA (Xing-Gang <i>et al.</i> , 2009)	1.91	Direct (Γ - Γ)
PBE-GGA (Luciana Fernández-Werner <i>et al.</i> , 2011)	2.10	Direct (Γ - Γ)		
N doped TiO <sub>2</sub> at O site	Present work	PBE-GGA+SOC	1.63	Direct (Γ - Γ)
N doped TiO <sub>2</sub> at O site		PBE-GGA+SOC	0.32	Indirect



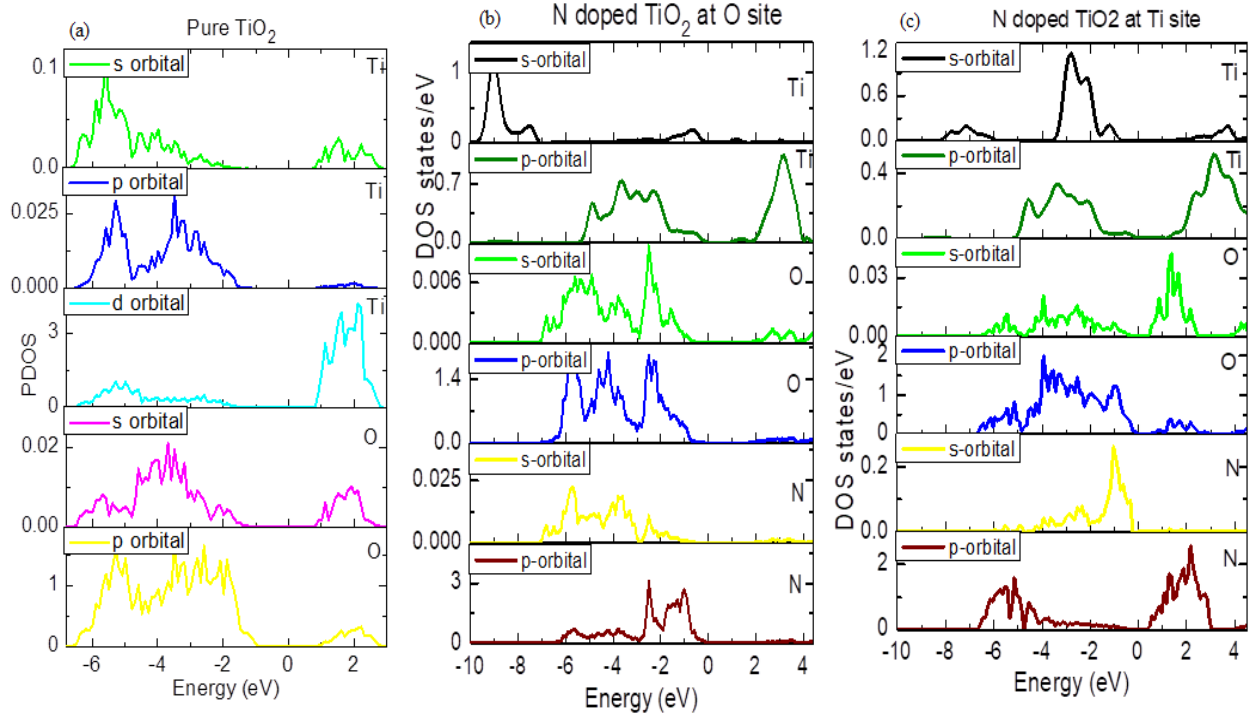


**Fig. 2:** Band structure of (a) Pure  $\text{TiO}_2$  (b) N doped  $\text{TiO}_2$  at O site and (c) N doped  $\text{TiO}_2$  at Ti site

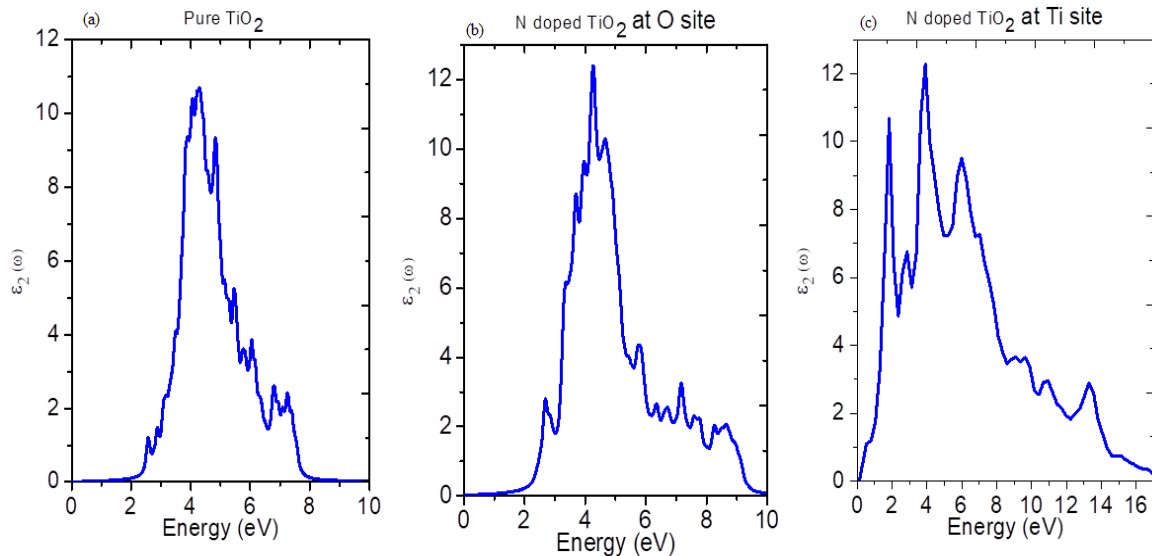


**Fig. 3:** Total density of states (DOS) of (a) pure  $\text{TiO}_2$  (b) N doped  $\text{TiO}_2$  at O site (c) N doped  $\text{TiO}_2$  at Ti site





**Fig. 4:** Total density of states (DOS) of (a) Pure TiO<sub>2</sub> (b) N doped TiO<sub>2</sub> at O site (c) N doped TiO<sub>2</sub> at Ti site



**Fig.**

**5:** The imaginary part of the frequency-dependent dielectric function of (a) Pure TiO<sub>2</sub> (b) N doped TiO<sub>2</sub> at O site and (c) N doped TiO<sub>2</sub> at Ti site

**3.3. Optical Properties of Pure TiO<sub>2</sub> and N Doped TiO<sub>2</sub>**

The study of optical properties of a material is crucial to have an insight on its characteristics for the applications in optoelectronic systems and devices. Normally, optical properties

describe the behavior of the material when exposed to electromagnetic emissions. From the literature, it was found that the exploration of the optical features relating to N-doped TiO<sub>2</sub> has not been done. The optical properties of pure TiO<sub>2</sub> and N doped TiO<sub>2</sub> are computed using



random phase approximation (RPA) based on PBE-GGA+SOC. The optical parameter considered in this paper is the imaginary part of the dielectric function  $\varepsilon_2(\omega)$ . The imaginary part  $\varepsilon_2(\omega)$  of frequency dependent of the dielectric function relates to how light is absorbed by the medium (LAWAL, 2017, Lawal *et al.*, 2017, Radzwan *et al.*, 2020, Radzwan *et al.*, 2018) The obtained imaginary part of frequency-dependent dielectric function of TiO<sub>2</sub>, N doped TiO<sub>2</sub> at O site, and N doped TiO<sub>2</sub> at Ti site are displayed in Figure 5 (a), (b) and (c) respectively. For undoped TiO<sub>2</sub> crystal, the first critical point sometimes called the edge of optical absorption (optical gap) occurred at about 2.30 eV, this value corresponds to the fundamental bandgap. The calculated optical gap of undoped TiO<sub>2</sub> is in good agreement with experimental data. This point split the valence band maximum and conduction band minimum. For N doped TiO<sub>2</sub> at O and Ti sites, the optical gap was found to be 1.63 and 0.32 eV. The optical gaps of 1.63 and 0.32 eV for N doped TiO<sub>2</sub> at O site and Ti site, suggest that N doped TiO<sub>2</sub> is a promising candidate for solar cells and near infrared opto-electronic applications such as gas sensing and optical communication.

#### 4.0 Conclusion

In this paper, optoelectronic properties of pure and N doped TiO<sub>2</sub> were investigated by using computational approaches within the framework of DFT. The calculations were carried out using Quantum Espresso and Yambo packages. To exploit the potential of the studied materials structural, electronic and optical properties were investigated. The generated structures of the studied materials were optimized. The results of the optimized structures concerning structural parameters were found to agree with the experimental results. The obtained bandgap energy for pure TiO<sub>2</sub> was found to 2.30 eV and this value is close to experimental results. Substituting N atom in the TiO<sub>2</sub> at the O and Ti sites led to the reduction in the energy bandgap and this trend is in good agreement with previous work. Our

findings confirmed that non-metal doping narrows the energy gap of semiconductor materials. The optical gap of 1.63 and 0.32 eV for N doped TiO<sub>2</sub> at O and Ti sites indicated that N-doped TiO<sub>2</sub> can be used to detect light in the near infrared region and visible light wavelengths. Direct gap, narrowing effects and strong light absorption of N-doped TiO<sub>2</sub> in the near infrared to visible light region suggest that the investigated material is suitable for solar cells and near infrared optoelectronic applications.

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#### Conflict of Interest

The authors declared no conflict of interest. This work was carried out in collaboration among all authors. All authors read and approved the final manuscript

