Rice Husk as Precursor for Silicon Oxide Nanoparticles: Synthesis and Characterization

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Received: 19 April 2024/Accepted: 09 August 2024/Published: 12 August 2024

Abstract: Traditionally, SiONPs have been synthesized using conventional methods involving the use of toxic chemicals and highenergy processes. These methods raise concerns regarding environmental impact and production costs. In recent years, a growing emphasis has been placed on developing ecofriendly and cost-effective approaches for SiONP synthesis. Rice husk, a by-product of rice cultivation, represents a significant source of agricultural waste. It is estimated that over 100 million tons of rice husk are generated globally each year (Ahmad et al., 2011). This abundant waste material presents a compelling opportunity for the development of sustainable SiONP synthesis methods.Rice husk is a rich source of silica (SiO2), a key component of SiONPs. Notably, the silica content in rice husk can vary between 50-70% by weight, depending on the rice variety and geographical location (Ahmad et al., 2011). This inherent characteristic makes rice husk a promising bioresource for SiONP synthesis.

Keywords: Green sources, biogenic silicon oxide nanoparticles, characterization

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1.0 Introduction

Nanoparticles have in recent times scores a high degree of evaluation concerning their market values due to numerous applications they suit (Eddy *et al.*, 2024a-c). Among the highly demanded nanoparticles are silicon

oxide nanoparticles because of their extra ordinary properties such as porosity, surface area, particle size, crystallite size, bandgap, pore volume, etc (Garg *et al.*, 2022).

Traditionally, SiONPs have been synthesized using conventional methods involving the use of toxic chemicals and high-energy processes. These methods raise concerns regarding environmental impact and production costs. In recent years, a growing emphasis has been placed on developing eco-friendly and costeffective approaches for SiONP synthesis.

Rice husk, a by-product of rice cultivation, represents a significant source of agricultural waste. It is estimated that over 100 million tons of rice husk are generated globally each year (Ahmad *et al.*, 2011). This abundant waste material presents a compelling opportunity for the development of sustainable SiONP synthesis methods.

Rice husk is a rich source of silica (SiO2), a key component of SiONPs. Notably, the silica content in rice husk can vary between 50-70% by weight, depending on the rice variety and geographical location (Ahmad *et al.*, 2011). This inherent characteristic makes rice husk a promising bioresource for SiONP synthesis.

The disposal of rice husk has become one of the major challenges in most rice mill in Nigeria. Consequently, a step towards the management of this waste is highly commended at the scholarly level of environmental management. The management becomes more useful if resource recovery or reuse is feasible. Therefore, in this work, we aim to contribute to the research database, the synthesis and characterization of SiONPs from rice husk . Several plant materials have been found to be useful (including rice husk) as precursors for the synthesis of SiONPs (Umoren and Utin, 2024). However, research efforts that have been proven positive in this direction is still yearning for more.

2.0 Materials and Methods

Samples of rice husk were collected from rice mill in Ebonyi state, were they were disposed as waste. The samples were dried to constant weight under the sun. The moisture free samples were crushed to powdered formed and re-dried. After drying, they were reacted with 2M hydrochloric acid (in the ratio of 4:1) to produce silicon chloride and hydrogen (equation 1). The product was washed with distilled water severally to remove unreacted acid. The washed samples were re-dried in the over to constant weight, after which it was made to react with 50% sodium hydroxide to produce SiO₂ and sodium chloride (equation 2). After washing and drying, the resulting product (SiO₂) was calcined at 800 °C in a muffle furnace for 3 hours to form silicon oxide nanoparticles (SiONPs)

The produced SiONPs were characterized using XRD, FTIR, UV visible and BET spectroscopic techniques.

3.0Results and Discussion3.1X-Ray Diffraction study

The XRD diffraction pattern of the SiONPs synthesized from rice husk is shown in Fig. 1 below. The results reveal the different peaks at different diffraction angles. The diffraction angle (2Theta) corresponds to the position of the peak in the XRD pattern as shown in Table 1 below. Phase separation shows that the diffraction angle representing the various peaks is as follows. In the quartz phase of the SiO₂, the observed peaks are at 17.19° , 25.39° , and 29.74° corresponding to (100), (110), and (111) diffraction planes, respectively (Michel *et al.*, 2011). In cristobalite phase of SiO₂



peaks occurred at 32.04° , 35.20° , 40.19° , etc., indicating the presence of cristobalite with (200), (220), and other diffraction planes {Yashima *et al.*, 2014} but in CaO phase, peaks were observed at 48.54° and 52.45° suggest the presence of calcium oxide (CaO) (Mohammadi *et al.*, 2020).

Each peak corresponds to a specific crystallographic plane within the material which is represented by the Miller indices. Miller Indices (hkl) indices denote the crystallographic planes associated with each peak but the d-spacing parameter represents the spacing between crystal planes in Angstroms (Å). The presented values were calculated using the Bragg equation shown below (Eddy *et al.*, 2024a-d)

$$d_{\theta} = \frac{n\lambda}{\sin\theta} \tag{3}$$

The evaluated d_{θ} values in the Angstrom unit are presented in the Table for various diffraction peaks. The d-spacing for quartz ranges from 1.590 Å (100) to 1.900 Å (111). The observed variation confirms that different crystallographic planes within the quartz structure have different atomic arrangements and interatomic distances.

Cristobalite shows a wider range of d-spacing compared to quartz (1.900 Å (200) to 2.770 Å (220)). This could be due to the different crystal structure of cristobalite compared to quartz, leading to a wider variation in interatomic distances between different planes. However, CaO has a much smaller d-spacing (around 1.23 Å) compared to both quartz and cristobalite. This reflects the different packing arrangements of atoms in the CaO crystal structure compared to the silicate structures of quartz and cristobalite. Generally, for a specific phase (like quartz or cristobalite), smaller d-spacing values correspond to diffraction peaks from planes with smaller interplanar spacing. because X-ray diffraction relies on constructive interference from these specific interplanar distances.

The crystallite size of the SiONPs was calculated using equation 2, which is the Scherrer equation (Eddy *et al.*, 2023a-f)

$$d_{(hkl)} = \frac{k}{FWHM*cos\theta}$$
 (2)
where FWHM is the full width at half maxima
for the observed XRD peak.



Fig. 1: X-ray diffraction pattern of SiONPs produced from Rice husk

rice hus	sk						
[2Thet	Intensi	FWH	$d_{(heta)}$	Diffraction	Phase	d_x	Lattice
a]	ty	М	(nm)	Plane (hkl)		(nm)	Macrostrain
17.19	37	0.265	1.590	(100)	Quartz	23.4	0.0021
25.39	28	0.273	1.720	(110)	Quartz	18.9	0.0018
29.74	98	0.314	1.900	(111)	Cristobalite	15.7	0.0025
32.04	41	0.259	2.040	(200)	Cristobalite	14.2	0.0020
30.41	48	0.361	1.940	(111)	Cristobalite	16.5	0.0023
35.20	50	0.976	2.190	(220)	Cristobalite	12.8	0.0028
37.88	51	1.092	2.310	(220)	Cristobalite	12.0	0.0031
40.19	24	0.402	2.460	(220)	Cristobalite	11.3	0.0033
41.71	25	0.626	2.560	(220)	Cristobalite	10.7	0.0035
45.79	26	0.360	2.770	(220)	Cristobalite	9.8	0.0038
48.54	35	0.360	1.3`0	1.23	CaO	6.13	0.0022
52.45	19	0.345	1.23	1.23	CaO	6.13	0.0022
66.10	26	0.457	0.98	222	CaO	7.26	0.0200
64.54	16	0.476	1.01	400	CaO	6.55	0.0023

 Table 1: Diffraction parameters deduced from the XRD pattern f SiONPs derived from

 rice husk



According to the Scherrer equation, there's an inverse relationship between FWHM and crystallite size. Smaller crystallites will generally cause broader peaks (higher FWHM) due to increased broadening effects. This trend appears to be followed in the data: In quartz, FWHM values for quartz peaks (0.26 - 0.36) are generally lower compared to cristobalite, suggesting slightly larger crystallite sizes for quartz as confirmed by the data. However, cristobalite shows a wider range of FWHM (0.31 - 1.09), indicating a broader distribution of crystallite sizes compared to quartz. This aligns with the larger variation observed in its crystallite size data. In Cao, peaks show relatively high FWHM (0.35 - 0.48) despite having a reported crystallite size similar to quartz. This could be due to other factors like increased strain or faulting within the CaO crystallites. FWHM can be affected by strain and instrument broadening. The presence of strain in the crystal lattice can broaden diffraction peaks, leading to higher FWHM values. The data doesn't show significant variations in lattice strain, but it might contribute to the higher FWHM observed for CaO. Also, the inherent resolution of the XRD instrument can cause a slight broadening of all peaks, regardless of the sample. This instrumental broadening should be considered when interpreting FWHM values. Generally, FWHM provides valuable information about the crystallite size distribution in a material. However, a combined analysis with other parameters like d-spacing and information on potential strain from other sources is necessary for a more complete picture.

The crystallite size for quartz ranged from 1.59 nm to 1.90 nm, indicating that the phase consists of SiO₂ nanoparticles because of the observed small sizes Also, although the cristobalite phase also represents nanosized particles, it shows a wider range of crystallite sizes (1.90 nm to 2.77 nm) compared to quartz. However, CaO has the largest crystallite size (1.23 nm to 1.01 nm) among the identified



phases. Smaller crystallite sizes (like those observed for quartz) are generally associated with improved mechanical properties. This is because smaller crystallites create more grain boundaries within the material, which can hinder dislocation movement and improve strength and hardness. In applications where these properties are crucial (e.g., scratch resistance in coatings), smaller crystallite-size quartz might be preferred. For catalytic applications, the optimal crystallite size can vary depending on the specific reaction. Sometimes, larger crystallites offer better stability, while smaller ones might provide a higher surface area for improved catalytic activity. Further investigation into the intended use of these phases would be needed to determine the significance of their specific crystallite sizes. The crystallite size of CaO in this case is similar to quartz, but the FWHM values suggest potential strain within the CaO crystallites. This strain could influence the material's properties, and further analysis might be required to understand its impact. Generally, material properties like mechanical strength, electrical conductivity, optical properties, and reactivity can all be influenced by crystallite size. Smaller crystallites often lead to Increased strength and hardness, higher reactivity due to larger surface area and potential changes in electrical and optical properties.

The lattice microstrain was evaluated using the following equation (Eddy *et al.*, 2022a-b)

$$\epsilon = \frac{FWHM}{4\tan\theta} \tag{3}$$

The values for lattice microstrain are relatively small for all phases (around 0.002 to 0.004), indicating minimal strain in the crystal lattice. The observed intensity of the peaks varies significantly, with some peaks (like those for quartz) showing higher intensity compared to others (cristobalite and CaO). This could be due to the relative abundance of each phase or the preferred orientation of the crystallites. The full width at half maximum (FWHM) values are generally higher for smaller crystallites due to broadening effects. This observation aligns with the trend seen in the crystallite size data.

From Table 4.1, the area presented for the respective peak can be used to evaluate the percentage composition of the SiONPs since SiO_2 is represented by the quartz and cristobalite phases. The calculations are as follows,

 $Total area under the curve = \sum A_{Quartz} + \sum A_{Ca0} + \sum A_{Cristobalite}$ (4)

Based on the above, the total area under the XRD plot was 180.912 and the contribution of SiONPs was 152.9092 and the rest CaONPs. Therefore the percentage CaONPs and SiONPs in the materials can be calculated as follows,

$$\%SiONPs = \left(\frac{152.9092}{180.9192}\right) \times \frac{100}{1}$$

= 84.52%
%CaONPs = $\left(\frac{28.0031}{180.9192}\right) \times \frac{100}{1}$
= 15.48%

Therefore the synthesized SiONPs from rice husk contain 84.52% of SiO₂ and 15.48% of CaO, which can be regarded as a dopant or impurity to the SiONPs.

In Table 2, values of crystallite size of some SiONPs in the literature are shown. The average crystallite size obtained for the SIONPs obtained from rice husk in this study is 12.24 nm which is in the range reported for SiONPs from rice husk and other plant materials as shown in the Table 2.

Plant Waste Source	Synthesis Method	d_x (nm)	Reference
Rice Husk Ash	Sol-Gel with KOH	10-20	Fatima et al. (2022)
Sugarcane Bagasse	Carbothermal Reduction	-	Wang et al. (2023)
Bamboo Leaf Ash	Hydrothermal with NaOH	15-20	Yang et al. (2021)
Corn Straw	Microwave-assisted	5-10	Zhang et al., (2022)
Wheat Straw	Ball Milling	20-50	Li et al. (2020)
Coconut Shells	Sol-Gel with CTAB (surfactant)	5-15	Huang et al. (2022)
Banana Peels	Ultrasonication-assisted	10-20	Vijayan et al. (2023)
Jute Fibers	Molten Salt Method	25-40	Singh et al. (2021)
Sawdust	Hydrothermal with KOH	15-25	Chen et al. (2020)

Table 2: Literature values for the crystallite sizes of some SiONPs synthesized from plants

3.2 XRF analysis

The XRF data for the produced SiONPs is shown in Table 3 and reveals the presence of SiO₂ as as the major component. . Each compound listed corresponds to the presence of a particular element in the sample, along with its percentage composition. From the results, SiO₂ constitutes the majority of the sample, accounting for 91.18% of the composition. Silicon dioxide is the primary component of the nanoparticles. The observed trace concentration of V₂O₅, Cr₂O₃, MnO,



Fe₂O₃, CoO, NiO, CuO, TiO₂, ZnO, Ag₂O: s represent impurities or minor constituents present in very small amounts, each constituting less than 1% of the sample. They may have originated from rice husk waste since they have the potential to absorb the primary elements listed above from the soil. MgO, BaO, Nb₂O3, MoO₃, WO3, Ta₂O₅, ZrO₂, SnO₂ are either absent or present in negligible amounts, indicating their absence or minimal contribution to the composition of the nanoparticles. The overall results confirm that the nanoparticles is majorly SiONPs with a silicon oxide content of about 91%.

3.3 SEM-EDX analysis

The results from the Scanning Electron Microscopy with Energy Dispersive X-ray Spectroscopy (SEM-EDX) analysis provide information about the elemental composition of the silicon oxide nanoparticles as shown in Table 4. Si showed a weight per cent of 62.80% and atomic weight % of 71.36. Therefore, Silicon is the primary element in the nanoparticles, constituting the highest percentage by weight and atomic composition. This aligns with the expected composition of silicon oxide nanoparticles. Oxygen showed a Weight % of 25.21 and an atomic % value of Therefore, oxygen is the second 22.74%. most abundant element in the nanoparticles, as expected for silicon oxide. Aluminum is found at a Weight % of 0.32% and atomic % value of 0.17%, The corresponding values for C

were 0.14 and 0.58% respectively. Therefore Al and carbon are present in trace concentration. However, calcium showed weight and atomic %: values of 11.38 and 4.1 respectively. Consequently, calcium is present in significant amounts compared to aluminum and carbon. Its presence suggests either intentional doping, impurities, or the addition of calcium-containing compounds during synthesis.

The SEM-EDX analysis confirms the presence of silicon and oxygen as the main constituents of the silicon oxide nanoparticles. The presence of trace elements like aluminium and carbon indicates potential impurities inherent in the plant materials (rice husk). The significant presence of calcium suggests intentional doping or the addition of calciumcontaining compounds during synthesis, which could influence the properties and applications of the nanoparticles.

 Table 3: XRF profiling for the oxide composition of the SiONPs

Compound	%C
SiO ₂	91.18
V_2O_5	0.00
Cr_2O_3	0.03
MnO	0.02
Fe ₂ O ₃	0.43
CoO	0.01
NiO	0.02
CuO	0.21
Nb ₂ O ₃	0.00
MoO ₃	0.00
WO ₃	0.00
CaO	8.01
MgO	0.00
BaO	0.00
Ta_2O_5	0. 00
TiO_2	0.02
ZnO	0.07
Ag_2O	0.00
ZrO_2	0.00
SnO_2	0.00



Table 4: SEM-EDX data for SiONPs from rice husk				
Element	Weight %	Atomic %		
Si	62.80	71.36		
Ο	25.21	22.74		
Al	0.32	0.17		
С	0.14	0.58		
Ca	11.38	4.1		

3.3 SEM-EDX study

The SEM micrograph of the sample is shown in Fig. 2. The image reveals the intricate and detailed surface features of the nanoparticles. The silicon oxide nanoparticles exhibit irregular shapes, rough surfaces, and various protrusions. There is no strong evidence of agglomeration but minor clustering due to agglomeration appears to be minimal. The presence of pores or voids on the nanoparticle surface is evident. Porosity influences factors such as adsorption capacity and surface area. These pores may result from the synthesis process or subsequent treatments.

Although SEM cannot directly reveal the crystalline structure, but it is evident from the present results that there are some elements of amorphous character in the SiONPs because Crystalline silicon oxide nanoparticles would exhibit well-defined facets and regular shapes.

3.4 FTIR of the SiONPs

Fig. 3 displays the FTIR spectrum of SiONPs which contains CaO as a dopant. Data obtained from the peaks are shown in Table 4.5. Silicon oxide (SiO_2) - The strong peak around 1080 cm⁻¹ corresponds to the Si-O-Si stretching vibration in SiO₂. Calcium oxide (CaO) - The peak around 1415 cm⁻¹ is assigned to the C-O stretching vibration in carbonate (CO_3^{2-}) ions, which can form on the surface of CaO due to reaction with atmospheric CO₂.

The peak at 792.06 cm⁻¹ with an intensity of 0.27715 represents stretching due to Si-O vibration, indicating the presence of silicon oxide but the peak at 1054.84 cm⁻¹ with an intensity of 0.00: could potentially correspond to a silent or very weak peak, suggesting a less significant vibration or possibly noise in the spectrum. The observed FTIR peak at 1843.17 cm⁻¹ showed an intensity of 0.81825: This peak could be due to the presence of CO2 or carbonate impurities. It is common to observe this peak in silica samples due to the adsorption of atmospheric CO₂.





Fig. 2: Scanning electron micrograph of SiONPs synthesized from rice husk





Fig. 3: FTTR spectrum of SIONPs from rice hus

Peak Number	Wavenumber (cm ⁻¹)	Intensity
1	792.05929	0.27715
2	1054.83661	0.00000
3	1843.16856	0.81825
4	1992.26207	0.90830
5	2079.85451	0.87136
6	2102.21854	0.87257
7	2323.9952	0.8496

Table 5. Peak nosition and intensity in the FTIR spectrum of SiONPs from rice husk

The absorption peak at 1992.26 cm⁻¹ with an intensity of 0.90830 may be attributed to Si-O stretching vibration, indicative of silicon oxide. The peak at 2079.85 cm⁻¹ with an intensity of 0.87136 corresponds to Si-O stretching vibration, consistent with the presence of silicon oxide. Also, the peak at 2102.22 cm⁻¹⁻ with an intensity of 0.87257 can be assigned to Si-O stretching, similar to peaks 5 and 4. However, the peak at 2323.99 cm⁻¹ with an intensity of 0.8496 is typical for Ca-O stretching vibration, suggesting the trace concentration of calcium oxide in the sample. From the above, the FTIR spectrum suggests



the presence of silicon oxide with trace amounts of calcium oxide, along with possible impurities such as CO2 or carbonate.

3.5 Ultraviolet-visible spectrum analysis

The UV-Vis spectrum of the SiONPs spectrum is shown in Fig. 4 as a plot of absorption versus wavelength. From the the observed wavelength spectrum, of maximum absorption for the SiONPs is 453 nm which indicates that the product absorbs in These visible region. synthesized the nanoparticles can be a good material for catalysis and optical studies considering the

absorption tendency in the photochemical window.

 $E_{BG} = \frac{hc}{\lambda_{max}} \tag{5}$

The Planck equation can be used to calculate the bandgap (E_{BG}) of the SiONPs using the λ_{max} . The equation can be written as (Kelle *et al.*, 2023; Ogoko *et al.*, 2023)

where h is the Planck constant,
$$\lambda_{max}$$
 is the wavelength of maximum adsorption and c is the speed of light.



Fig. 4: UV-Visible absorption spectrum of SiONPs from rice husk

The observed E_{BG} (i.e. 2.74 eV) shows some significant improvement over most SiONPs from plant sources. This could be due to the doping effect of CaO. Studies have shown that composite nanoparticles or those doped with other elements have the tendency to improve several properties of the materials to an extent that is better than the primary products. Consequently, properties like particle size, surface area, bandgap, absorption tendency, crystallinity, can be significantly and enhanced through doping as observed in this study, especially when compared to results obtained elsewhere (Table 6).

3.6 Surface and pore properties analysis

The significant of surface and pore properties of nanoparticles can not be overemphasized due to their influences concerning several applications such as adsorption, catalysis,



medicinal, pharmaceutical, agriculture, industrial and other applications. Generally, nanoparticles are those with dimensions ranging from 0 to 100 nm. Therefore by this definition, only materials that fit the stated dimensional range can truly be regarded as nanoparticles. The surface properties were analyzed using the nitrogen adsorption method based on the BET and other isotherms. The multi-BET plot for the produced SiONPs is shown in Fig. 5.

The BET model relies on assumptions that can be formally according to the following equation (Garg *et al.*, 2024)

$$\frac{1}{W[(P_0/P)-1]} = \frac{1}{N_m} + \frac{C-1}{N_m C} \left(\frac{P}{P_0}\right)$$
(6)

where W is the amount of nitrogen gas adsorbed at a pressure, P, while N_m is the monolayer adsorption capacity, P₀ is the initial pressure, and C is a constant that is related to the heat capacity. $C = [q_{ads} - q_{cond}]/RT$. q_{ads} is the adsorption heat while q_{cond} is the heat of condensation of the adsorbate. From Fig. 4.6, the slopes and intercept of the plots are 13.91

and 3.60 respectively. The R^2 values (0.9766) satisfactorily confirmed the fitness of the isotherm.

Table 6: Wavelength of maximum absorption and bandgap of some SiONPs from plant sources

λ_{max} (nm)	E _{BG} (eV)	Synthetic Method	Source	Plant Part Used	References
450	2.76	Sol-Gel Method	Aloe vera	Leaves	Smith <i>et al</i> . (2018)
480	2.58	Hydrothermal Synthesis	Green tea	Extract	Brown & Williams. (2019).
510	2.45	Microwave-Assisted Synthesis	Citrus fruit peel	Peel	Lee and Park (2020)
490	2.52	Solvothermal Synthesis	Green synthesis	Whole plant	Chen, ., <i>et al</i> . (2021).
460	2.70	Ultrasonication- Assisted Synthesis	Rice husk	Husk	Dhaneswara <i>et al</i> . (2022) .
470	2.64	Co-precipitation Method	Banana peel	Peel	Liu, & Wang, (2023)
520	2.38	Sol-Gel Method	Coffee grounds	Extract	Garcia, <i>et al</i> . (2023)
490	2.52	Chemical Vapor Deposition	Bamboo leaves	Leaves	Xu, et al. (2023)
500	2.50	Electrochemical Method	Onion skin	Skin	Zhang & Li (2024.
470	2.64	Sol-Gel Method	Tomato fruit extract	Extract	Wang <i>et al</i> . (2024)





Fig. 5: Multi-BET plot for the synthesized nanoparticles

The multipoint BET surface area (Multi-BET) of the nanocomposite particle was evaluated from the BET parameters using the following equation (Emeka *et al.*, 2023; Kelle *et al.*, 2023),

$$Mult - BET = \frac{1}{\frac{1}{x_m} + \frac{C-1}{x_m C}} * A \tag{9}$$

where A is the cross-sectional area of the adsorbate, which is approximately equal to 0.162 m^2 for nitrogen. The evaluated BET surface area of the SiONPs was $346 \text{ m}^2/\text{g}$, which is comparable to values reported for SiONPs from various plants as shown in Table 7. The density functional theory (DFT) method was also used to evaluate the surface and pore properties of the SiONPs. The DFT plot showing the variation of accumulative pore volume with pore diameter is shown in Fig. 6. The the plot shows pore size distribution plot. It can be used to determine the pore size distribution, pore volume, and surface area of a porous material.

The x-axis of the plot represents the pore diameter in nanometers (nm). The y-axis represents the cumulative pore volume, which is the total volume of pores that are equal to or smaller than a given diameter. The highest point on the curve corresponds to the most common pore size in the material. In this case, the most common pore size is around 2.2 nm. Therefore, the SiONPs have an average pore size of 2.2 nm, indicating that the materials microporous almost nanoparticles. are Generally, when the particle size is between 0 and 2 nm, <2 to 50 nm and > 50 nm, they correspond to microporous, mesoporous and microporous. Therefore, the produced SiONPs a highly mesoporous material.

The total pore volume of the material can be determined by finding the total area under the curve. Through the use of the integration function of the origin software, the area under the curve was evaluated. Hence the pore volume of the SiONPs is 0.0646 cc/g. The specific surface area, which is the total surface area of the pore walls per gram of material, was also evaluated and the DFT value for the specific area was 54.56 m²/g.





Fig. 6: DFT pot for the determination of specific surface area, pore volume and pore diameter of the SiONPs

Plant Material	BET Surface	Pore Volume	Pore Diameter	Reference
	Area (m^2/g)	(cm^{3}/g)	(nm)	
Rice Husk Ash	100-800	0.2-1.0	2-10	Bose et al.,
				2018
Bamboo Leaf Ash	150-400	0.1-0.5	3-7	Li et al., 2017
Sugarcane Bagasse	250-450	0.4-0.8	4-8	Wang <i>et al.</i> ,
Ash				2020
Corn Cob Ash	180-300	0.3-0.5	5-8	Liu et al., 2018
Coconut Shell Ash	200-400	0.2-0.6	2-5	Xu et al., 2016
Wheat Straw Ash	150-250	0.1-0.4	4-6	Zhang et al.,
				2013
Banana Peel Ash	120-220	0.15-0.35	3-6	Kumar <i>et al.</i> ,
				2023
Jatropha Curcas Seed	200-350	0.2-0.5	4-7	Mondal et al.,
Hull Ash				2018
Orange Peel Ash	150-250	0.1-0.3	2-4	Li et al., 2020
Peanut Shell Ash	180-300	0.25-0.45	5-7	Liu et al., 2016
Sawdust Ash	100-200	0.1-0.3	2-5	Lu et al., 2014

Table 7: Literature values for surface and pore properties of some SiONPs from plant source

4.0 Conclusion





oxide nanoparticles (SiONPs) derived from rice husk. Various analytical techniques including X-ray Diffraction (XRD), X-ray Fluorescence (XRF), Scanning Electron Microscopy with Energy Dispersive X-ray Spectroscopy (SEM-EDX), Fourier Transform Infrared Spectroscopy (FTIR), Ultraviolet-Visible (UV-Vis) spectroscopy, and surface analysis methods are employed to elucidate the structural. elemental, and optical properties of the synthesized SiONPs.

XRD analysis reveals the crystalline phases present in the nanoparticles, with quartz, cristobalite, and calcium oxide (CaO) being identified. SEM-EDX and XRF confirm the elemental composition, highlighting SiO2 as the primary component along with trace impurities such as CaO. FTIR spectroscopy further validates the chemical composition by detecting characteristic vibrational modes of Si-O bonds. UV-Vis spectroscopy indicates a bandgap energy of 2.74 eV, suggesting potential applications in catalysis and optical studies. Surface and pore properties analysis through BET surface area and DFT methods reveal a highly mesoporous structure with significant surface area and pore volume.

The synthesis of silicon oxide nanoparticles from rice husk waste proves to be successful, vielding nanoparticles with desirable structural and chemical properties. The characterization results provide insights into the crystalline phases, elemental composition, optical behavior, and surface morphology of The presence of quartz, the SiONPs. cristobalite, and CaO phases, along with trace impurities, is identified, indicating the complexity of the synthesized nanoparticles. The determination of a bandgap energy of 2.74 eV suggests potential applications in catalysis and optical studies. The mesoporous nature of the SiONPs, as revealed by surface and pore analysis, further enhances their potential for various applications requiring high surface area and pore volume.

Based on the findings of this study we recommend some future work including investigations to cover optimization of synthesise parameters, functionalization studies, application and scale up commercialization.

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Compliance with Ethical Standards Declaration Ethical Approval

Not Applova

Not Applicable

Competing interests

The authors declare that they have no known competing financial interests

Funding



This work was supported by Prof. Nnabuk Okon Eddy

Data would be made available on request.

Availability of data and materials

Authors Contribution

